# Comparative Study on Polyimides from 3,3'- and 4,4'-Linked Diphthalic Anhydride 

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#### Abstract

SYNOPSIS 1,4-Bis(2,3-dicarboxyphenoxy)benzene dianhydride, 1,4-bis(3,4-dicarboxyphenoxy)benzene dianhydride, bis(2,3-dicarboxyphenoxy)sulfide dianhydride, bis(3,4-dicarboxyphenoxy)sulfide dianhydride, and $2,3,3^{\prime}, 4^{\prime}$-tetracarboxy diphenyl sulfide dianhydride were synthesized from 3 -chlorophthalic anhydride and 4 -chlorophthalic anhydride. Bis(2,3dicarboxyphenyl)sulfone and bis(3,4-dicarboxyphenyl)sulfone were obtained by the oxidation of the corresponding bis(dicarboxyphenyl)sulfide by hydrogen peroxide. The polyimides from the dianhydrides mentioned above and $4,4^{\prime}$-oxydianiline were prepared. The properties, such as dynamic mechanical behavior, thermooxidative stability, stress-strain behavior, chemical resistance, and permeability to some gases have been in investigated for the isomeric polyimides. © 1996 John Wiley \& Sons, Inc.


## INTRODUCTION

Most aromatic polyimides from bridged dianhydride studied so far were prepared from 4,4'-linked diphthalic anhydride, such as $3,3^{\prime}, 4,4^{\prime}$-benzophenone dianhydride, $3,3^{\prime}, 4,4^{\prime}$-oxydiphthalic anhydride, 2,2-bis (3,4-dicarboxyphenyl) hexafluoropropane dianhydride, etc. Only a few works have dealt with $3,3^{\prime}$-linked diphthalic anhydride-based polyimides. ${ }^{1-3}$

Chlorophthalic anhydride was chosen as a common starting material in our laboratory for the preparation of a series dianhydride as shown in Scheme 1. Chlorophthalic anhydride was synthesized from $o$-xylene which was chlorinated first, followed by the oxidation catalyzed by vanadium oxide and the cocatalysts in the gas phase (see Scheme $2)$. In this case, the chlorophthalic anhydride was a mixture of isomers, i.e., 3 -chlorophthalic anhydride and 4 -chlorophthalic anhydride. The ratio of these two isomers was about 1:2.

In the present work, we report a comparative

[^0]study on the polyimides based on the $3,3^{\prime}$-linked and $4,4^{\prime}$-linked diphthalic anhydride. The properties, such as dynamic mechanical behavior, glass transition temperature, thermooxidative stability, stress-strain behavior, chemical resistance, and permeability to some gases for these isomeric polyimides were investigated.


Scheme 1 Dianhydride from chlorophthalic anhydride.


Scheme 2 Synthesis of chlorophthalic anhydride.

## EXPERIMENTAL

## Materials

3-Chlorophthalic anhydride and 4-chlorophthalic anhydride were obtained by distillation of crude chlorophthalic anhydride. The purity of these two compounds was about $97 \%$ determined by gas chromatography. The boiling point of 3- and 4-chlorophthalic anhydride was determined at 750 mmHg

Table I Dianhydrides and Their Melting Points
(3, Absreviation

Table II The Elemental Analysis of New Dianhydrides

| Dianhydride | Elemental Analysis |  |  |  |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | Founded |  |  | Calculated |  |  |
|  | C | H | S | C | H | S |
| $\left(3,3^{\prime}\right) \mathrm{HQDPA}$ | 65.70 | 2.62 | - | 65.67 | 2.49 | - |
| ( $3,3^{\prime}$ ) TDPA | 59.00 | 1.87 | 10.15 | 58.89 | 1.84 | 9.82 |
| $\left(3,4^{\prime}\right.$ )TDPA | 59.46 | 1.75 | 10.57 | 58.89 | 1.84 | 9.82 |
| $\left(3,3^{\prime}\right)$ SDPA | 53.41 | 1.52 | 8.83 | 53.63 | 1.67 | 8.93 |

for the samples in $99 \%$ purity to be 322 and $298^{\circ} \mathrm{C}$, respectively.

1,4-Bis (3,4-dicarboxyphenoxy) benzene dianhydride $\left[\left(4,4^{\prime}\right) \mathrm{HQDPA}\right]$ and 1,4-bis (2,3-dicarboxyphenoxy) benzene dianhydride [ $\left(3,3^{\prime}\right) \mathrm{HQDPA}$ ] were prepared from the corresponding $N$-methyl chlorophthalimide and hydroquinone in DMAc in the presence of potassium carbonate at $150^{\circ} \mathrm{C}$ for 8 h , followed by the hydrolysis of the diimide in aqueous sodium hydroxide under reflex for 8 h . After acidation with dilute sulfuric acid, the crude tetraacid was filtered and boiled in $50 \%$ sulfuric acid for 30 min , washed with deionized water six times, and dehydrated at $250^{\circ} \mathrm{C}$ for 2 h to obtain the dianhydride. The total yield was about $75 \%$.

Diphenylthioether dianhydrides (TDPA) were prepared from the corresponding $N$-methyl chlorophthaliimide, sulfur, and potassium carbonate in DMAc according to the procedure reported in Ref.
4. Diphenylsolfone dianhydrides (SDPA) were prepared by the oxidation of TDPA with hydrogen peroxide. ${ }^{5}$

## Polymerization

Equivalent dianhydride and diamine was reacted in DMAc to make a solid content of $15 \%$ at room temperature. To the obtained viscous solution was added 2.5 equivalent acetic anhydride, the stirring was continued for 1 h , then 0.2 equivalent triethylamine was added as the catalyst for the imidization. The polyimide was formed as the powder in about 2 h ; the equal volume ethanol was poured into the mixture. After stirring for 0.5 h , the polyimide powder was filtered and washed in hot ethanol three times, then dried and heated at $280^{\circ} \mathrm{C}$. For the soluble polyimide, such as $\left(3,3^{\prime}\right)$ TDPA/ $\left(4,4^{\prime}\right)$-oxydianiline (ODA), the polymer solution was poured into ethanol; after washing and drying, the polymer was heated at $280^{\circ} \mathrm{C}$ for 1 h .

## Chemical Resistance

Polyimide film about $20 \mu \mathrm{~m}$ thick was immersed in chemicals at room temperature for 30 days, then washed with ethanol and water and dried in air for testing.

## Gas Permeability

For the apparatus and procedure for determining the permeability, see the literature. ${ }^{6}$


Figure 1 Dynamic mechanical analysis of polyimides based on HQDPA/ODA.


Figure 2 Dynamic mechanical analysis of polyimides based on TDPA/ODA.

## Measurement

Dynamic mechanical analysis was taken in a DuPont 982 instrument with a heating rate of $5^{\circ} \mathrm{C} /$ min . The specimens were about 2 mm thick and compression-molded from the resin powder at $370^{\circ} \mathrm{C}$ under 10 MPa .

The tensile measurements were carried out on an Instron Model 1122 testing machine at room temperature. Crosshead speed was $2.0 \mathrm{~cm} / \mathrm{min}$, corresponding to the relative strain rate of $1.0 / \mathrm{min}$. Thermal analysis was taken in a DELTA TGA 7 thermogravimetric system in air under the heating rate of $10^{\circ} \mathrm{C} / \mathrm{min}$.

## RESULTS AND DISCUSSION

## Synthesis

The melting points of the dianhydride used in this work are listed in Table I. The elemental analysis
of the newly prepared dianhydrides are listed in Table II.
( $3,3^{\prime}$ ) TDPA, $\left(3,4^{\prime}\right)$ TDPA, and ( $3,3^{\prime}$ ) SDPA were quite soluble in water compared with the corresponding $4,4^{\prime}$-linked analogs, so that the extraction of ether from the aqueous solution of the tetraacid may be necessary for recovering the product. The polyimide from ( $3,3^{\prime}$ ) SDPA was too brittle to be cast as a film and has too high a softening temperature to be compression-molded, so, unfortunately, there was no mechanical property that could be measured in this work.

## Dynamic Mechanical Behavior

As can be seen in Figures 1 and 2, the (4,4) polyimides have a broad $T_{\beta}$ transition, but one very weak for ( $3,3^{\prime}$ ) polymers. The $T_{\beta}$ might be a common characteristic for the ( $4,4^{\prime}$ ) polyimides that we investigated, such as $\left(4,4^{\prime}\right) \mathrm{ODPA} / \mathrm{ODA}$,

Table III $T_{g}$ and $T_{\beta}$ of Polyimides

| Polyimide | $T_{g} \mathrm{~K}\left({ }^{\circ} \mathrm{C}\right)$ | $T_{\beta}(\mathrm{K})$ | $T_{\beta}(\mathrm{K}) / T_{g}(\mathrm{~K})$ |
| :---: | :---: | :---: | :---: |
| $\left(3,3^{\prime}\right)$ TDPA/ODA | 561 (288) | - | - |
| $\left(4,4^{\prime}\right)$ TDPA/ODA | 540 (267) | 411 | 0.76 |
| $\left(3,4^{\prime}\right)$ TDPA/ODA | 539 (266) | 412 | 0.76 |
| $\left(3,3^{\prime}\right) \mathrm{HQDPA} / \mathrm{ODA}$ | 542 (269) | - | - |
| $(4,4) \mathrm{HQDPA} / \mathrm{ODA}$ | 518 (245) | 390 | 0.75 |
| $\left(4,4^{\prime}\right)$ SDPA/ODA | 608 (335) | 444 | 0.73 |
| $\left(4,4^{\prime}\right)$ TDPA/MDA | 545 (272) | 401 | 0.74 |
| (4,4') HQDPA/MDA | 544 (271) | 405 | 0.75 |
| $\left(4,4^{\prime}\right) \mathrm{ODPA} / \mathrm{MDA}$ | 552 (279) | 404 | 0.73 |

(4,4') ODPA/(4,4')-methylenedianiline (MDA), and ( $4,4^{\prime}$ ) HQDPA/MDA as well, except the polyimide from $3,3^{\prime}, 4,4^{\prime}$-biphenyl dianhydride, which will be reported elsewhere.

Furthermore, the glass transition temperature for ( $3,3^{\prime}$ ) TDPA/ODA and ( $3,3^{\prime}$ ) HQDPA/ODA is about $20^{\circ} \mathrm{C}$ higher than that for the corresponding $4,4^{\prime}$-linked polymers. Nevertheless, the ( $3,4^{\prime}$ ) TDAP/ODA has a similar behavior as that for ( $4,4^{\prime}$ ) TDPA /ODA. The absence of $T_{\beta}$ for $\left(3,3^{\prime}\right)$ linked polyimides may result from the steric hindrance of the rotation around the bond to the imide group vicinally located. ${ }^{7}$

The $T_{g}$ and $T_{\beta}$ of the polyimides studied are shown in Table III. The ratio of $T_{\beta}$ to $T_{g}$ in Kelvin scale are constant around 0.75 .

## Thermooxidative Stability

From Table IV, we can see that the thermooxidative stability of polyimides is independent of the isomerism. The polymers from TDPA/ODA have the highest stability, but that from SDPA/ODA, the lowest among the studied polyimides.

## Stress-Strain Curves

The polyimide from ( $3,3^{\prime}$ ) TDPA/ODA appears brittle, but the ( $4,4^{\prime}$ ) analog is strong and tough (Fig. 3). Both polymers from ( $3,3^{\prime}$ ) HQDPA / ODA and (4,4') HQDPA / ODA are strong and tough (Fig. 4).

This behavior may be caused by the rigidity of the polymer chain. As mentioned above, in ( $3,3^{\prime}$ ) TDPA-based polyimide, the rotation of imide was restricted by the adjacent $\mathrm{S}-\mathrm{C}$ bond to the imide group, whereas in the ( $3,3^{\prime}$ ) HQDPA-based polymer, the $-\mathrm{OC}_{6} \mathrm{H}_{4} \mathrm{O}$ - linkage acts as a buffer for the restriction of the rotation of the imide ring, so that the rigidity of the chain is decreased.

Table IV The Thermooxidative Stability of Polyimides

| Polyimide | $T_{c}\left({ }^{\circ} \mathrm{C}\right)$ | $T_{5 \%}\left({ }^{\circ} \mathrm{C}\right)$ |
| :--- | :---: | :---: |
| $\left(3,3^{\prime}\right)$ TDPA/ODA | 530 | 556 |
| (4,4 $\left.4^{\prime}\right)$ TDPA/ODA | 535 | 559 |
| $\left(3,4^{\prime}\right)$ TDPA/ODA | 526 | 571 |
| (3,3)HQDPA/ODA | 505 | 531 |
| (4,4)HQDPA/ODA | 503 | 536 |
| (3,3')SDPA/ODA | 483 | 521 |
| $\left(4,4^{\prime}\right)$ SDPA/ODA | 448 | 522 |
| $\left(4,4^{\prime}\right)$ ODPA/ODA | 480 | 546 |

[^1]

Figure 3 Stress-strain curves of polythioether-imides (a) $\left(3,3^{\prime}\right)$ TDPA/ODA; (b) $\left(4,4^{\prime}\right)$ TDAP/ODA

## Chemical Resistance (Tables V and VI)

Ignoring the $\pm 5 \%$ change from the value of block samples for the experimental error which may caused by the determination and/or the preparation of the sample, we can see from the results that

1. Polyimides based on HQDPA/ODA have better chemical resistance than those based on TDPA/ODA. The latter lost the yield point in most chemicals.
2. ( $4,4^{\prime}$ ) Polyimides are quite resistant to chloroform, but the ( $3,3^{\prime}$ ) counterpart became swollen or even dissolved in that. Compared with Kapton, the polyimide from pyromellitic dianhydride and ODA, these polymers, especially the HQDPA-based ones, are quite stable in $10 \%$ aqueous sodium hydroxide.


Figure 4 Stress-strain curves of polyetherimides: (a) $\left(3,3^{\prime}\right) \mathrm{HQDPA} / O D A ;(b)\left(4,4^{\prime}\right) \mathrm{HQDPA} / O D A$.
Table V Chemical Resistance of Polyetherimides ${ }^{\text {a }}$

| Medium | $\left(3,3^{\prime}\right) \mathrm{HQDPA} / \mathrm{ODA}$ |  |  |  |  | $\left(4,4^{\prime}\right) \mathrm{HQDPA} / \mathrm{ODA}$ |  |  |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | Yield |  |  | Break |  | Yield |  |  | Break |  |
|  | Modulus (MPa) | Tensile <br> Strength (MPa) | Elongation (\%) | Tensile Strength (MPa) | Elongation (\%) | Modulus (MPa) | Tensile Strength (MPa) | Elongation (\%) | Tensile Strength (MPa) | Elongation (\%) |
| Control | 1891 | 114 | 7.5 | 116 | 128.5 | 1484 | 104 | 12.2 | 128 | 84.1 |
| $10 \% \mathrm{H}_{2} \mathrm{SO}_{4}$ | $\begin{aligned} & 1697 \\ & (-10.3) \end{aligned}$ |  |  | $\begin{gathered} 99 \\ (-15.1) \end{gathered}$ | $\begin{gathered} 7.6 \\ (-94.1) \end{gathered}$ | $\begin{aligned} & 1512 \\ & \quad(2.0) \end{aligned}$ | 94 | 14.1 | 124 | 84.3 |
| $10 \% \mathrm{HCl}$ | $\begin{aligned} & 1807 \\ & (-4.0) \end{aligned}$ | $\begin{gathered} 99 \\ (-13.0) \end{gathered}$ | $\begin{gathered} 6.7 \\ (-10.4) \end{gathered}$ | $\begin{aligned} & 111 \\ & (-4.2) \end{aligned}$ | $\begin{aligned} & 126 \\ & (-2.1) \end{aligned}$ | $\begin{array}{r} 1626 \\ (9.4) \end{array}$ | $\begin{gathered} 97 \\ (-6.6) \end{gathered}$ | $\begin{gathered} 13.7 \\ (12.3) \end{gathered}$ | $\begin{aligned} & 123 \\ & (-4.4) \end{aligned}$ | $\begin{gathered} 80.1 \\ (-0.2) \end{gathered}$ |
| $10 \% \mathrm{HNO}_{3}$ | $\begin{gathered} 1816 \\ (-4.0) \end{gathered}$ | $\begin{aligned} & 103 \\ & (-9.8) \end{aligned}$ | $\begin{gathered} 7.4 \\ (-1.3) \end{gathered}$ | $\begin{gathered} 94 \\ (-19.4) \end{gathered}$ | $\begin{gathered} 10.0 \\ (-92.2) \end{gathered}$ | 1594 (7.4) | $\begin{gathered} 98 \\ (-6.5) \end{gathered}$ | $\begin{gathered} 13.7 \\ (12.3) \end{gathered}$ | $\begin{aligned} & 130 \\ & (1.3) \end{aligned}$ | $\begin{gathered} 81.2 \\ (-3.4) \end{gathered}$ |
| 10\% HF | $\begin{aligned} & 1752 \\ & (-7.4) \end{aligned}$ |  |  | $\begin{gathered} 100 \\ (-14.0) \end{gathered}$ | $\begin{gathered} 7.7 \\ (-94.0) \end{gathered}$ | $\begin{aligned} & 1855 \\ & (25.0) \end{aligned}$ |  |  | $\begin{aligned} & 124 \\ & (-3.7) \end{aligned}$ | $\begin{gathered} 12.4 \\ (-85.3) \end{gathered}$ |
| $10 \% \mathrm{NaOH}$ | $\begin{gathered} 1795 \\ (-5.1) \end{gathered}$ | $\begin{gathered} 99.7 \\ (-12.3) \end{gathered}$ | $\begin{array}{r} 6.9 \\ (-8.0) \end{array}$ | $\begin{gathered} 84 \\ (-27.7) \end{gathered}$ | $\begin{gathered} 15.9 \\ (-87.6) \end{gathered}$ | $\begin{aligned} & 1652 \\ & (11.4) \end{aligned}$ | $\begin{gathered} 95 \\ (-8.9) \end{gathered}$ | $\begin{gathered} 14.0 \\ (14.5) \end{gathered}$ | $\begin{aligned} & 125 \\ & (-2.4) \end{aligned}$ | $\begin{aligned} & 85 \\ & (0.6) \end{aligned}$ |
| $30 \% \mathrm{H}_{2} \mathrm{O}_{2}$ | $\begin{aligned} & 1748 \\ & (-7.6) \end{aligned}$ | $\begin{gathered} 101 \\ (-11.2) \end{gathered}$ | $\begin{gathered} 7.7 \\ (2.7) \end{gathered}$ | $\begin{gathered} 96 \\ (-17.3) \end{gathered}$ | $\begin{gathered} 45.3 \\ (-64.7) \end{gathered}$ | $\begin{aligned} & 1629 \\ & (9.8) \end{aligned}$ | $\begin{gathered} 93 \\ (-10.6) \end{gathered}$ | $\begin{gathered} 13.7 \\ (12.3) \end{gathered}$ | $\begin{aligned} & 130 \\ & (1.1) \end{aligned}$ | $\begin{gathered} 88.9 \\ (5.7) \end{gathered}$ |
| Ethanol | $\begin{aligned} & 1680 \\ & (-11.2) \end{aligned}$ | $\begin{array}{r} 76.5 \\ (-32.7) \end{array}$ | $\begin{array}{r} 9.4 \\ (25.3) \end{array}$ | $\begin{gathered} 98 \\ (-15.8) \end{gathered}$ | $\begin{aligned} & 117 \\ & (-8.0) \end{aligned}$ | $\begin{aligned} & 1733 \\ & (16.8) \end{aligned}$ | $\begin{gathered} 91 \\ (-12.5) \end{gathered}$ | $\begin{gathered} 15.0 \\ (23.0) \end{gathered}$ | $\begin{aligned} & 137 \\ & (6.5) \end{aligned}$ | $\begin{gathered} 94.1 \\ (11.8) \end{gathered}$ |
| Acetone | $\begin{aligned} & 1608 \\ & (-15.0) \end{aligned}$ | $\begin{gathered} 65.9 \\ (-42.0) \end{gathered}$ | $\begin{gathered} 4.5 \\ (-40.0) \end{gathered}$ | $\begin{gathered} 89 \\ (-23.0) \end{gathered}$ | $\begin{aligned} & 131 \\ & (2.2) \end{aligned}$ | $\begin{array}{r} 1545 \\ (4.1) \end{array}$ | $\begin{gathered} 83 \\ (-19.7) \end{gathered}$ | $\begin{gathered} 6.3 \\ (-48.4) \end{gathered}$ | $\begin{aligned} & 120 \\ & (-6.9) \end{aligned}$ | $\begin{gathered} 85.2 \\ (2.5) \end{gathered}$ |
| Chloroform |  |  |  |  |  | $\begin{aligned} & 1689 \\ & (13.8) \end{aligned}$ | $\begin{gathered} 96 \\ (-7.5) \end{gathered}$ | $\begin{array}{r} 13.7 \\ (12.3) \end{array}$ | $\begin{aligned} & 140 \\ & (8.9) \end{aligned}$ | $\begin{gathered} 88.9 \\ (5.8) \end{gathered}$ |
| Toluene | $\begin{aligned} & 1846 \\ & (-2.4) \end{aligned}$ | $\begin{gathered} 97.7 \\ (-14.1) \end{gathered}$ | $\begin{gathered} 6.8 \\ (-9.3) \end{gathered}$ | $\begin{gathered} 92.5 \\ (-20.3) \end{gathered}$ | $\begin{gathered} 43.6 \\ (-66.1) \end{gathered}$ | $\begin{aligned} & 1665 \\ & (12.2) \end{aligned}$ | $\begin{aligned} & 105 \\ & (1.0) \end{aligned}$ | $\begin{gathered} 11.9 \\ (-2.5) \end{gathered}$ | $\begin{aligned} & 140 \\ & (9.1) \end{aligned}$ | $\begin{aligned} & 88.1 \\ & (4.8) \end{aligned}$ |
| Glacial acetic acid | $\begin{aligned} & 1613 \\ & (-14.7) \end{aligned}$ | $\begin{gathered} 64 \\ (-43.6) \end{gathered}$ | $\begin{gathered} 6.5 \\ (-13.3) \end{gathered}$ | $\begin{gathered} 80.7 \\ (-30.5) \end{gathered}$ | $\begin{gathered} 95.7 \\ (-25.5) \end{gathered}$ | $\begin{aligned} & 1936 \\ & (30.5) \end{aligned}$ | $\begin{aligned} & 103 \\ & (-1.2) \end{aligned}$ | $\begin{gathered} 10.9 \\ (-10.7) \end{gathered}$ | $\begin{aligned} & 155 \\ & (20.6) \end{aligned}$ | $\begin{gathered} 72.3 \\ (-14.0) \end{gathered}$ |
| Ethyl acetate | $\begin{aligned} & 1591 \\ & (-15.9) \end{aligned}$ | $\begin{gathered} 61 \\ (-46.1) \end{gathered}$ | $\begin{gathered} 3.5 \\ (-53.3) \end{gathered}$ | $\begin{gathered} 77.2 \\ (-33.5) \end{gathered}$ | $\begin{gathered} 98.2 \\ (-23.6) \end{gathered}$ | $\begin{aligned} & 1645 \\ & \quad(10.8) \end{aligned}$ | $\begin{gathered} 90 \\ (-13.6) \end{gathered}$ | $\begin{gathered} 14.2 \\ (16.4) \end{gathered}$ | $\begin{aligned} & 127 \\ & (-1.3) \end{aligned}$ | $\begin{gathered} 76.3 \\ (-9.3) \end{gathered}$ |
| Petroleum ether | $\begin{aligned} & 1672 \\ & (-11.6) \end{aligned}$ | $\begin{gathered} 92 \\ (-18.8) \end{gathered}$ | $\begin{gathered} 4.5 \\ (-40.0) \end{gathered}$ | $\begin{aligned} & 112 \\ & (-3.7) \end{aligned}$ | $\begin{gathered} 130.9 \\ (1.9) \end{gathered}$ | $\begin{aligned} & 1793 \\ & (20.8) \end{aligned}$ | $\begin{aligned} & 113 \\ & (9.1) \end{aligned}$ | $\begin{gathered} 14.4 \\ (18.0) \end{gathered}$ | $\begin{aligned} & 139 \\ & (8.2) \end{aligned}$ | $\begin{gathered} 71.6 \\ (-14.9) \end{gathered}$ |
| Mineral oil | $\begin{aligned} & 1624 \\ & (-14.1) \end{aligned}$ | $\begin{gathered} 96 \\ (-15.6) \end{gathered}$ | $\begin{gathered} 7.2 \\ (-4.0) \end{gathered}$ | $\begin{gathered} 93 \\ (-20.0) \end{gathered}$ | $\begin{gathered} 42.1 \\ (-67.2) \end{gathered}$ | $\begin{aligned} & 2000 \\ & (34.8) \end{aligned}$ | $\begin{aligned} & 103 \\ & (-1.2) \end{aligned}$ | $\begin{aligned} & 12.4 \\ & (1.6) \end{aligned}$ | $\begin{aligned} & 137 \\ & (6.7) \end{aligned}$ | $\begin{gathered} 84.9 \\ (1.0) \end{gathered}$ |

[^2]Table VI Chemical Resistance of Polythioetherimides ${ }^{\text {a }}$

| Medium | (3,3')TDPA/ODA |  |  |  |  | (4,4')TDPA/ODA |  |  |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | Yield |  |  | Break |  | Yield |  |  | Break |  |
|  | Modulus (MPa) | Tensile Strength (MPa) | Elongation (\%) | Tensile Strength (MPa) | Elongation (\%) | Modulus (MPa) | Tensile Strength (MPa) | Elongation (\%) | Tensile Strength (MPa) | Elongation (\%) |
| Control | 1999 |  |  | 107 | 6.5 | 1776 | 121 | 15.0 | 129 | 64.7 |
| $10 \% \mathrm{H}_{2} \mathrm{SO}_{4}$ | $\begin{aligned} & 1840 \\ & (-7.9) \end{aligned}$ |  |  | $\begin{gathered} 81.2 \\ (-24.3) \end{gathered}$ | $\begin{gathered} 5.1 \\ (-21.5) \end{gathered}$ | $\begin{aligned} & 1549 \\ & (-12.8) \end{aligned}$ |  |  | $\begin{gathered} 100 \\ (-22.7) \end{gathered}$ | $\begin{gathered} 10.0 \\ (-84.5) \end{gathered}$ |
| $10 \% \mathrm{HCl}$ | $\begin{aligned} & 1845 \\ & (-7.7) \end{aligned}$ | 102 | 6.4 | $\begin{gathered} 99 \\ (-7.3) \end{gathered}$ | $\begin{gathered} 6.8 \\ (4.6) \end{gathered}$ | $\begin{gathered} 1714 \\ (-3.4) \end{gathered}$ | $\begin{gathered} 106 \\ (-12.7) \end{gathered}$ | $\begin{array}{r} 12.5 \\ (-16.7) \end{array}$ | $\begin{gathered} 102 \\ (-21.1) \end{gathered}$ | $\begin{gathered} 28.3 \\ (-56.3) \end{gathered}$ |
| $10 \% \mathrm{HNO}_{3}$ | $\begin{aligned} & 1945 \\ & (-2.7) \end{aligned}$ |  |  | $\begin{aligned} & 104 \\ & (-2.6) \end{aligned}$ | $\begin{gathered} 6.3 \\ (-3.1) \end{gathered}$ | $\begin{aligned} & 1637 \\ & (-7.8) \end{aligned}$ |  |  | $\begin{gathered} 105 \\ (-18.8) \end{gathered}$ | $\begin{gathered} 28.5 \\ (-56.4) \end{gathered}$ |
| 10\% HF | $\begin{aligned} & 1712 \\ & (-14.3) \end{aligned}$ |  |  | $\begin{gathered} 92 \\ (-14.6) \end{gathered}$ | $\begin{gathered} 5.2 \\ (-20.0) \end{gathered}$ | $\begin{aligned} & 1898 \\ & \quad(6.8) \end{aligned}$ |  |  | $\begin{aligned} & 117 \\ & (-9.5) \end{aligned}$ | $\begin{gathered} 10.4 \\ (-84.2) \end{gathered}$ |
| $10 \% \mathrm{NaOH}$ |  |  |  |  |  | $\begin{aligned} & 1581 \\ & (-11.0) \end{aligned}$ |  |  | $\begin{gathered} 98.7 \\ (-23.7) \end{gathered}$ | $\begin{gathered} 12.0 \\ (-81.5) \end{gathered}$ |
| $30 \% \mathrm{H}_{2} \mathrm{O}_{2}$ | $\begin{aligned} & 1810 \\ & (-9.4) \end{aligned}$ | 96 | 6.6 | $\begin{gathered} 92 \\ (-13.8) \end{gathered}$ | $\begin{gathered} 29.0 \\ (346.2) \end{gathered}$ | $\begin{gathered} 1670 \\ (-6.0) \end{gathered}$ |  |  | $\begin{gathered} 111 \\ (-13.9) \end{gathered}$ | $\begin{gathered} 13.6 \\ (-79.0) \end{gathered}$ |
| Ethanol | $\begin{aligned} & 1712 \\ & (-14.3) \end{aligned}$ | 82 | 7.9 | $\begin{gathered} 85 \\ (-20.7) \end{gathered}$ | $\begin{gathered} 26.1 \\ (301.5) \end{gathered}$ | $\begin{gathered} 1666 \\ (-6.3) \end{gathered}$ |  |  | $\begin{gathered} 102 \\ (-21.2) \end{gathered}$ | $\begin{gathered} 10.2 \\ (-84.2) \end{gathered}$ |
| Acetone | $\begin{aligned} & 1646 \\ & (-17.6) \end{aligned}$ | 73 | 7.5 | $\begin{gathered} 82 \\ (-23.2) \end{gathered}$ | $\begin{gathered} 122.6 \\ (1786) \end{gathered}$ | $1784$ (0.5) | $\begin{gathered} 91 \\ (-24.8) \end{gathered}$ | $\begin{gathered} 8.6 \\ (-42.7) \end{gathered}$ | $\begin{gathered} 108 \\ (-16.2) \end{gathered}$ | $\begin{gathered} 49.5 \\ (-23.5) \end{gathered}$ |
| Chloroform |  |  |  |  |  | $\begin{aligned} & 1700 \\ & (-4.3) \end{aligned}$ | $\begin{aligned} & 111 \\ & (-7.9) \end{aligned}$ | $\begin{gathered} 13.9 \\ (-6.7) \end{gathered}$ | $\begin{gathered} 108 \\ (-16.2) \end{gathered}$ | $\begin{gathered} 40.0 \\ (-38.5) \end{gathered}$ |
| Toluene | $\begin{aligned} & 1781 \\ & (-10.9) \end{aligned}$ |  |  | $\begin{gathered} 98 \\ (-8.4) \end{gathered}$ | $\begin{gathered} 7.4 \\ (13.8) \end{gathered}$ | $\begin{aligned} & 1734 \\ & \quad(0.5) \end{aligned}$ |  |  | $\begin{aligned} & 124 \\ & (-4.9) \end{aligned}$ | $\begin{gathered} 12.4 \\ (-80.8) \end{gathered}$ |
| Glacial acetic acid | $\begin{aligned} & 1700 \\ & (-14.9) \end{aligned}$ | 77 | 6.2 | $\begin{gathered} 71 \\ (-33.8) \end{gathered}$ | $\begin{gathered} 57.2 \\ (780) \end{gathered}$ | $\begin{aligned} & 1918 \\ & (8.0) \end{aligned}$ |  |  | $\begin{aligned} & 117 \\ & (-9.8) \end{aligned}$ | $\begin{gathered} 12.6 \\ (-80.5) \end{gathered}$ |
| Ethyl acetate | $\begin{aligned} & 1617 \\ & (-19.1) \end{aligned}$ | 74 | 5.8 | $\begin{gathered} 71 \\ (-33.6) \end{gathered}$ | $\begin{gathered} 13.1 \\ (101.5) \end{gathered}$ | 2043 <br> (15.0) |  |  | $\begin{aligned} & 123 \\ & (-4.9) \end{aligned}$ | $\begin{gathered} 13.2 \\ (-79.6) \end{gathered}$ |
| Petroleum ether | $\begin{aligned} & 1920 \\ & (-3.9) \end{aligned}$ |  |  | $\begin{aligned} & 108 \\ & (0.7) \end{aligned}$ | $\begin{gathered} 6.6 \\ (1.5) \end{gathered}$ | 1860 (4.7) | $\begin{aligned} & 126 \\ & (4.2) \end{aligned}$ | $\begin{aligned} & 15.2 \\ & (1.2) \end{aligned}$ | $\begin{gathered} 134 \\ (3.9) \end{gathered}$ | $\begin{gathered} 60.0 \\ (-7.2) \end{gathered}$ |
| Mineral oil | $\begin{aligned} & 1898 \\ & (-5.1) \end{aligned}$ |  |  | $\begin{gathered} 109 \\ (1.9) \end{gathered}$ | $\begin{gathered} 7.2 \\ (10.8) \end{gathered}$ | $\begin{aligned} & 2373 \\ & \quad(33.6) \end{aligned}$ |  |  | $\begin{aligned} & 145 \\ & (12.0) \end{aligned}$ | $\begin{gathered} 12.1 \\ (81.3) \end{gathered}$ |

[^3]Table VII Permeability and Permselectivity of Polyimide

| Polyimide | $\mathrm{P}_{\mathrm{H}_{2}}$ |  |  | $\alpha_{\mathrm{H}_{2} / \mathrm{N}_{2}}$ |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | $30^{\circ} \mathrm{C}$ | $100^{\circ} \mathrm{C}$ | $150^{\circ} \mathrm{C}$ | $30^{\circ} \mathrm{C}$ | $100^{\circ} \mathrm{C}$ | $150^{\circ} \mathrm{C}$ |
| $\left(3,3^{\prime}\right) \mathrm{HQDPA} / O D A$ | 6.99 | 16.7 (2.39) | 26.1 (3.73) | 108 | 69.4 (0.64) | 55.3 (0.51) |
| (4,4')HQDPA/ODA | 4.00 | 11.5 (2.87) | 19.8 (4.95) | 118 | 73.0 (0.62) | 61.9 (0.52) |
| $\left(3,3^{\prime}\right) \mathrm{TDPA} / \mathrm{ODA}$ | 6.91 | 16.3 (2.36) | 24.6 (3.56) | 149 | 89.6 (0.60) | 70.3 (0.47) |
| $\left(4,4^{\prime}\right)$ TDPA/ODA | 4.68 | 13.2 (2.82) | 22.4 (4.79) | 221 | 111 (0.50) | 79.3 (0.36) |
|  | $\mathrm{P}_{\mathrm{O}_{2}}$ |  |  | $\alpha_{\mathrm{O}_{2} / \mathrm{N}_{2}}$ |  |  |
| (3, ${ }^{\prime}$ ) HQDPA/ODA | 0.416 | 1.22 (2.92) | 2.18 (5.19) | 6.3 | 5.1 (0.81) | 4.5 (0.71) |
| (4,4')HQDPA/ODA | 0.216 | 0.777 (3.55) | 1.50 (6.82) | 6.4 | 5.0 (0.78) | 4.7 (0.73) |
| $\left(3,3^{\prime}\right) \mathrm{TDPA} / \mathrm{ODA}$ | 0.346 | 0.988 (2.83) | 1.68 (4.80) | 7.5 | 5.5 (0.73) | 4.8 (0.64) |
| (4,4)TDPA/ODA | 0.198 | 0.661 (3.30) | 1.24 (6.26) | 10.1 | 6.3 (0.62) | 5.0 (0.50) |

$P: 10^{-10} \mathrm{~cm}^{3}(\mathrm{STP}) \mathrm{cm} / \mathrm{cm}^{2} \mathrm{~S} \mathrm{cmHg}$. The data in parentheses are the ratios to the values determined at $30^{\circ} \mathrm{C}$.

## The Permeability and Permselectivity for Gases

Polyimide homogeneous membranes have been tested for hydrogen, oxygen, and nitrogen. The results are compiled in Table VII. 4, $4^{\prime}$-Linked polyimides have higher permeability but lower permselectivity than have $3,3^{\prime}$-linked polyimides. This phenomenon may be caused by the difference in the free volume (see Table VIII), which was estimated according to the report by Lee. ${ }^{8}$ The temperature dependence of the permeability and permselectivity for $3,3^{\prime}$-linked polyimides is smaller than that for 4,4'-linked ones, except that the permselectivity for HQDPA-based polyimides is almost independent of temperature. On balance, the $3,3^{\prime}$-linked polyimide may be the more promising membrane material for gas separation.

Table VIII The Free Volume of Polyimides

| Polyimide | $V_{f}\left(\mathrm{~cm}^{3} / \mathrm{mol}\right)$ |
| :---: | :---: |
| $\left(3,3^{\prime}\right) \mathrm{HQDPA} / \mathrm{ODA}$ | 0.093 |
| $\left(4,4^{\prime}\right) \mathrm{HQDPA} /$ ODA | 0.088 |
| $\left(3,3^{\prime}\right) \mathrm{TDPA} /$ ODA | 0.097 |
| $\left(4,4^{\prime}\right) \mathrm{TDPA} /$ ODA | 0.092 |

## CONCLUSION

Isomeric polyimides can be synthesized by a similar procedure using 3 -chlorophthalic anhydride and 4chlorophthalic anhydride as the starting material. But the properties for these isomers are quite different. The 3, $3^{\prime}$-linked polyimides may be developed as novel materials for various applications.

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Received June 7, 1995
Accepted August 3, 1995


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    Journal of Applied Polymer Science, Vol. 59, 923-930 (1996)
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    CCC 0021-8995/96/060923-08

[^1]:    $T_{c}$ : Commence temperature of the weight loss $\left({ }^{\circ} \mathrm{C}\right) . T_{5 \%}$ : The temperature for $5 \%$ weight loss ( ${ }^{\circ} \mathrm{C}$ ).

[^2]:    ${ }^{\text {a }}$ Films about $20 \mu \mathrm{~m}$ thick were immersed at room temperature for 30 days.

[^3]:    ${ }^{\text {a }}$ Films about $20 \mu \mathrm{~m}$ thick were immersed at room temperature for 30 days.

